

Optical investigations of Ag/CeO² nanocomposites

T. Kim¹ , J. Heon¹ , Y. C. Seon1, , Dong-Soo Kim² , Won-Chun Oh³*

¹Department of Materials Science, Korea Aerospace University, Goyang, 10540, Republic of Korea 2 Dept. of Civil and Environmental Eng., Korea Advanced Institute of Science and Technology, South Korea

³Department of Advanced Materials Science & Engineering, Hanseo University, Seosan-si,

Chungcheongnam-do 31962, South Korea

*) Email: ycseon@kau.ac.kr

Received 30/6/2021, *Accepted,* 30/12/2021, *Published* 15/1/2022

Silver/ceria (Ag/CeO₂) nanocomposites were pre-pared from Ce(NO₃)₃ 6H₂O, AgNO₃, and NH₄OH with different molar ratios through a hydrothermal process, and thenwere completed by carrying out the precursors calcining at 750 \degree C for 2 h under air atmosphere. Below 1 % of Ag con-centration in Ag/CeO² nanocomposites, the Ag crystalline structure does not appear. XRD and TEM results show evidence of two different effects (the agglomeration and the barrier effects) governing the process of crystal growth. HR- TEM image and EDX elemental analysis of the $Ag/CeO₂$ nanocomposite confirmed that isolated Ag nanocrystals are dispersed in the $CeO₂$ matrix. The red shifts are attributed to the quantum confinement effect and the valence change of the $Ce⁺$ ions. Ag nanoparticles can help to improve the absorption of visible light, enhancing the absorption intensity of Ag/CeO² nanocomposite. These results are of great significance for the control of microstructure, crystallinity, and applications for the development of nanocomposite materials.

Keywords: Oxides; Optical; Composite.

1. INTRODUCTION

Nanocomposite materials show many special physical and chemical properties [1]. The $Ag/CeO₂$ nanocomposite attracted the attention of many experts and scholars because of the two elements of an Ag/CeO₂ system; the CeO₂ has a high oxygen storage capacity and anti-ultraviolet ray ability [2], which is applied in many areas, such as a polishing agent [3], catalyst [4], ultraviolet (UV) resistant [5], or gas sensors [6]. The other element is Ag, nanoparticles of Ag have been the key to many

important applications in catalysis [7] and optics [8]. Ag nanoparticles can also be used as an antimicrobial agent in various biomedical applications [9]. In addition, Ag is attractive for industrial applications due to its low cost and easy preparation.

Ag/CeO² nanocomposites are synthesized, besides having a two-element respective nature, and also extend the unique properties of composite materials to improve on its efficiency in optical, electrical, and chemical reactions. So, many techniques have been proposed for the synthesis of nanosized Ag/CeO² nanocomposite with promising control of properties, such as the coprecipitation method [10], the combustion method [7, 11], the impregnation method [12], the photochemical approach [13], the deposition precipitation method [14], the hydrothermal process [15], and the wet impregnation technique [16]. Synthesis of nanocomposite materials can be accomplished in a variety of ways, each method having favorable conditions and specificity. When using the hydrothermal process to synthesize nanocomposite powder, the main superiorities are simplicity of operation, low cost, good crystalline, ease of control, and the high reproducibility. $Ag/CeO₂$ has been reported for use as catalyst and in solid-oxide fuel cells $[11, 14]$, but for the Ag content of Ag/CeO₂, their nanocomposite structure and op- tical properties are rarely discussed.

The study of nanocomposite material has been of great interest from both fundamental and practical point of view.The physical properties of such nanocomposite materials can be combined to produce material of desired response. In order to further improve the properties of $CeO₂$, various Ag concentrations in the molar ratio range of $0-10\%$ on CeO₂ were used to synthesize Ag/CeO₂ nanocomposite was re- ported. The aim of this present work is to study the morphology and crystal structure of compositionally homogeneous $Ag/CeO₂$ nanocomposite synthesized by the hydrothermal method. The optical properties of the samples were also investigated.

2. EXPERIMENTAL

analytical grades. The Ag/CeO₂ nanocomposites were prepared by the hydrothermal method. We Cerium nitrate hexahydrate (Ce(NO₃)₃·6H₂O, 99.5 %, Aldrich Chemicals, USA), silver nitric $(AgNO₃)$, nitric acid $(HNO₃)$, polyethylene glycol (PEG 600), ammo- nia $(NH₄OH)$, methylene blue (MB), anhydrous ethanol (99.9 %, Fisher Chemicals), distilled water and alcohol, all at mixed aqueous solutions of Ce(NO₃)₃ 6H₂O and AgNO₃ together with different molar ratios, stirred about 30 min and then slowly dropped about 25 ml of NH4OH and 30 ml of PEG into the mixed solutions, respectively. The pH values were adjusted to 7.0, the above mixtures were poured into Teflon-lined stainless-steel autoclaves after vigorously stirring for about 2 h, and the autoclaves were maintained at 180 °C for 4 h, then air cooled to room temperature. The resultant precipitates were collected by centrifugation and washed with dis-tilled water and anhydrous alcohol three times. After these treatments, the resulting mixture was loaded into the oven at 80 °C for one day to gain precursors. At last, the precursors were calcined at 750 \degree C for 2 h to obtain Ag/CeO₂ nanocomposites. To determine the crystallite sizes and phases of the $Ag/CeO₂$ nanocomposites, the powders were identified by X-ray diffraction (XRD, Rigaku D/max-IV) using CuK*α* radiation (wavelength 1.5406 Å). The operation voltage and current were maintained at 40 kV and 20 mA, respectively. A scan rate of $2°/\text{min}$ was applied to record the patterns in the 2θ range of $25-65°$. Transmission electron microscope (TEM) and selected area electron diffraction (SAED) images were obtained on a (JEOL-2100) microscope with an accelerating voltage of 200 kV. The samples were dispersed in ethanol by ultrasonic treatment for about 30 min and thenthe suspension was dripped onto a copper grid for TEM. Semiquantitative microprobe analysis was performed with scanning electron microscopy (SEM, JEOL JSM-6500FE). Raman microspectroscopy, based on scattering, (ULVAC, laser 633 nm) was used to characterize the surface defects of samples. The UV–Vis diffuse reflectance spectra (UV– Vis DRS) was recorded in air at room temperature in the wavelength range

of 250–800 nm using a spectrophotometer (JASCO, V-760 Diffuse Reflectance Model).

Figure 1 XRD patterns of (**a**) pure CeO2, (**b**) 1 % Ag/CeO2, (**c**) 2.5 % Ag/CeO2, (**d**) 5 % Ag/CeO2 , (**e**) 10 % Ag/CeO2 .

3. RESULTS AND DISCUSSION

Figure [1 s](#page-1-0)hows the XRD patterns of pure $CeO₂$ and various molar ratios of Ag/CeO₂ nanocomposites produced by hydrothermal interaction. When the Ag concentration is lowerthan 1 %, only the cubic structure of $CeO₂$ appears in the XRD analysis because the Ag content is very low. All peaksdemonstrated that the main body is CeO2, with peak position of approximately 2*θ* 28.54°, 33.07°, 47.48°, 56.33°, and 59.08°, corresponding to a crystalline phase of (111),

(200), (220), (211), and (222), respectively; it is a face- centered cubic fluorite structure (JCPDS File No. 81-0792). The Ag concentration is lower than 1 %; only the cubic structure of $CeO₂$ appears in the XRD pattern. When the Ag concentration rises up to 2.5 %, the Ag crystalline phase begins to appear, metal Ag with a face-centered cubic phasestructure (JCPDS File No. 89-3722) was clearly observed, (111) , (200) , and (220) with the corresponding 2θ values of $38.12°$, $44.31°$, and 64.46°, respectively. The as-products size was estimated from line broadening of CeO₂ (111) peak using Scherrer's equation [17]:

$$
D = k\lambda/\beta \cos \theta) \tag{1}
$$

$$
(1)
$$

radiation, β the peak width at half-height and θ the value of the diffracted angle. By fitting various where *D* the average size of the crystallites, *k* Scherrer constant (0.98), λ the wavelength of peaks of the profiles to this formula, we obtained values in the range of 30 46 nm for the particle diameter.

The results are listed in Table 1. From those results, we distinguish three domains for $Ag/CeO₂$ nanocomposites, corresponding to three modes of the $CeO₂$ crystal size change. The first domain corresponds to the Ag adding con- centration between 0 and 2.5 %. In this domain, the $CeO₂$ mean crystallite size increases linearly with the Ag adding concentration, because of the agglomerate effects more than the barrier effect. The second domain corresponds to the Ag adding concentration between 2.5 and 5 %. In this domain, the increasing of the $CeO₂$ crystallite size becomes less pronounced, both agglomeration and barrier effect on growth trend to balance. The third domain corresponds to the Ag adding concentration between 5 and 10 %. In this domain, the $CeO₂$ mean crystallite size decreases with the Ag addingconcentration, the main reason is the Ag barrier affect the crystal growth, and led to the grain size reduction. The lattice parameters have changed with the adding Ag contents.

4. CONCLUSIONS

Ag/CeO² nanocomposites can be successfully synthesized by a hydrothermal method with different molar ratio con- centrations (0∼10 %) of Ag. XRD and SAED indicated that the particles of the Ag/CeO₂ were well crystalline, and HR-TEM as well as EDX elemental analysis proved the formation of nanocomposite between $CeO₂$ and Ag. Adding Ag will affect the structures and optical properties of $Ag/CeO₂$ nanocomposites. The agglomeration effect and the barrier effect governing the process of crystal growth. The crys-tal growth of agglomerated $CeO₂$ is faster than pure $CeO₂$ through high temperature heat treatment. The absorption in-tensity of $Ag/CeO₂$ nanocomposite increases in the visible region as the amount of Ag increases.

References

- [1] A.P. Alivisatos, Science 271 (1996) 933
- [2] X.L. Zhang, K.J. Klabunde, Inorg. Chem. 31 (1992) 1706
- [3] V.D. Kosynkin, A.A. Arzgatkina, E.N. Ivanov, M.G. Chtoutsa, A.I. Grabko, A.V. Kardapolov, N.A. Sysina, J. Alloys Compd. 303 (2000) 421
- [4] Liqaa H. Alwaan, Layla A. Jubur, Doaa A. Hussein, Exp. Theo. NANOTECHNOLOGY 4 (2020) 35
- [5] S. Tsunekawa, T. Fukuda, A. Kasuya, J. Appl. Phys. 87 (2000) 1318
- [6] T.S. Stefanik, H.L. Tuller, J. Eur. Ceram. Soc. 21 (2001) 1967
- [7] P.R. Sarode, K.R. Priolkar, P. Bera, M.S. Hegde, S. Emura, R. Ku- mashiro, Mater. Res. Bull. 37 (2002) 1679
- [8] I.A. Wani, A. Ganguly, J. Ahmed, T. Ahmad, Mater. Lett. 65 (2011) 520
- [9] M. Strathmann, J. Wingender, Int. J. Antimicrob. Agents 24 (2004) 234
- [10] S. Imamura, H. Yamada, K. Utani, Appl. Catal. A, Gen. 192 (2000) 221
- [11] P. Bera, K.C. Patil, M.S. Hegde, Phys. Chem. Chem. Phys. 2 (2000) 3715
- [12] Halima Mazari, Kheira Ameur, Aicha Boumesjed, Reski Khelifi, Sedik Mansouri, Nadia Benseddik, Nawal Benyahya, Zineb Benamara, Jean-Marie Bluet, Exp. Theo. NANOTECHNOLOGY 4 (2020) 49
- [13] S. Scire, C. Crisafulli, S. Giuffrida, C. Mazza, P.M. Riccobene, A. Pistone, G. Ventimiglia, C. Bongiorno, C. Spinella, Appl. Catal. A, Gen. 367 (2009) 138
- [14] J.H. Wang, M.L. Liu, M.C. Lin, Solid State Ion. 177 (2006) 939
- [15] Bassam Ramadhn Sarheed, Muhammed Abdul gafor, Mustafa. R. Al-Shaheen, Mohammed R. Alshaheen, Exp. Theo. NANOTECHNOLOGY 4 (2020) 59
- [16] M. Itome, A.E. Nelson, Catal. Lett. 106 (2006) 21

[17] B.D. Cullity, S.R. Stock, Elements of X-Ray Diffraction Prentice Hall, Upper Saddle River (2001)

© 2022 The Authors. Published by IFIA [\(https://etnano.com/\)](https://etnano.com/). This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution license [\(http://creativecommons.org/licenses/by/4.0/\)](http://creativecommons.org/licenses/by/4.0/).